



Microwave dielectric properties of $(1-x)\text{ZnMoO}_4-x\text{TiO}_2$ composite ceramics

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ARTICLE INFO

Article history:

Received 9 November 2010
Received in revised form 23 February 2011
Accepted 27 February 2011
Available online 5 March 2011

Keywords:

Microwave dielectric properties
Composite
LTCC
 $\text{ZnMoO}_4\text{-TiO}_2$

ABSTRACT

$(1-x)\text{ZnMoO}_4-x\text{TiO}_2$ ($x=0.0, 0.05, 0.158, 0.25, \text{ and } 0.35$) composite ceramics were synthesized by the conventional solid state reaction process. The sintering behavior, phase composition, chemical compatibility with silver, and microwave dielectric properties were investigated. All the specimens can be well densified below 950°C . From the X-ray diffraction analysis, it indicates that the triclinic wolframite ZnMoO_4 phase coexists with the tetragonal rutile TiO_2 phase, and it is easy for silver to react with ZnMoO_4 to form $\text{Ag}_2\text{Zn}_2(\text{MoO}_4)_3$ phase and hard to react with TiO_2 . When the volume fraction of TiO_2 (x value) increasing from 0 to 0.35, the microwave dielectric permittivity of the $(1-x)\text{ZnMoO}_4-x\text{TiO}_2$ composite ceramics increases from 8.0 to 25.2, the Q_f value changes in the range of 32,300–43,300 GHz, and the temperature coefficient τ_f value varies from -128.9 to 157.4 ppm/ $^\circ\text{C}$. At $x=0.158$, the mixture exhibits good microwave dielectric properties with a $\varepsilon_r=13.9$, a $Q_f=40,400$ GHz, and a $\tau_f=+2.0$ ppm/ $^\circ\text{C}$.

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1. Introduction

With the explosive growth of high frequency wireless communication technology, electronic components and substrates are in urgent need of improving performances and realizing miniaturization and integration. It is possible to integrate the passive components to a function module by low temperature cofired ceramic (LTCC) technology. Materials with low sintering temperature ($<960^\circ\text{C}$), high quality factor (Q_f) value and near zero temperature coefficient of resonant frequency (τ_f) are necessary for the applications [1–4]. Recently, a number of new microwave dielectric ceramics have been developed, such as AWO_4 and AMoO_4 ($A=\text{Mg, Mn, Zn, Ca, Sr, and Ba}$). AWO_4 compounds have good dielectric properties: permittivity is 8–17, Q_f value is 32,000–69,000 GHz, and τ_f value is -53 to -78 ppm/ $^\circ\text{C}$ [5]. AMoO_4 compounds are also suitable for applications of microwave dielectric materials, with low permittivity (7–11), low dielectric loss (37,000–90,000 GHz) and relatively small temperature coefficient of resonant frequency (-57 to -87 ppm/ $^\circ\text{C}$) [6]. Compared with AWO_4 , AMoO_4 ceramics have lower sintering temperature, especially ZnMoO_4 . However, all of them have negative τ_f values. Normally, there are two methods to design a material with a stable temperature coefficient: (a) composite materials by mixing two or more component materials with opposite τ_f values, including $\text{ZnAl}_2\text{O}_4\text{-TiO}_2$, $\text{Ca}_2\text{P}_2\text{O}_7\text{-TiO}_2$, $\text{CaWO}_4\text{-TiO}_2$, $\text{Zn}_2\text{Te}_3\text{O}_8\text{-TiO}_2$, $\text{Zn}_2\text{TiO}_4\text{-TiO}_2$, $\text{NiNb}_2\text{O}_6\text{-TiO}_2$, and $\text{Bi}_2\text{MoO}_6\text{-TiO}_2$ [7–13]; (b) formation of solid solutions, such as complex perovskites and

other systems [14–17]. In the recent investigation, ZnMoO_4 sintered at 800°C presents excellent microwave dielectric properties with $\varepsilon_r=8.67$ (permittivity), $Q_f=49,900$ GHz, $\tau_f=-87.49$ ppm/ $^\circ\text{C}$ [6]. In order to compensate the negative τ_f value of ZnMoO_4 , rutile TiO_2 ($\varepsilon_r=105$, $Q_f=46,000$ GHz, $\tau_f=+465$ ppm/ $^\circ\text{C}$) [18] was selected to form a $(1-x)\text{ZnMoO}_4-x\text{TiO}_2$ mixture. In this work, the synthesis, sintering behavior, phase composition, chemical compatibility with silver, and microwave dielectric properties of the $(1-x)\text{ZnMoO}_4-x\text{TiO}_2$ mixture ceramics were studied.

2. Experimental

All the samples were prepared by the conventional solid state reaction process. ZnO ($>99\%$, Sinopharm Chemical Reagent Co., Ltd., China), MoO_3 ($>99\%$, Fuchen Chemical Reagents, Tianjin, China) and rutile TiO_2 ($>99\%$, Linghua Co., Ltd., Zhaoqing, China) were used as starting materials. The ZnMoO_4 powders were synthesized by calcining in air at 600°C for 4 h. The ZnMoO_4 and TiO_2 powders were mixed according to the following stoichiometrics: $(1-x)\text{ZnMoO}_4-x\text{TiO}_2$ ($x=0.0, 0.05, 0.158, 0.25$ and 0.35 ; ZM, ZMT1, ZMT2, ZMT3, ZMT4 were used for abbreviations). The mixed powders were milled with ethanol and Zirconia milling media (2 mm in diameter) for 4 h using a planet ball-milling system whose running speed is 150 rpm, and then dried. The final powders mixed with PVA binder were pressed into cylinder samples (10 mm in diameter and 4–5 mm in height) and these pellets were sintered in the temperature range from 750°C to 975°C for 2 h with a heating rate of $3^\circ\text{C}/\text{min}$.

The phase compositions of the specimens were identified using X-ray diffraction with $\text{Cu K}\alpha$ radiation (Rigaku D/MAX-2400 X-ray diffractometry, Tokyo, Japan). The microstructure was observed with a scanning electron microscope (JSM-6460, JEOL, Tokyo, Japan). The bulk densities of the specimens were measured by Archimedes' method. The microwave dielectric properties were measured using the TE_{018} method with a network analyzer (8720ES, Agilent, Palo Alto, CA) and a temperature chamber (Delta 9023, Delta Design, Poway, CA). The temperature coefficient of resonant frequency (τ_f) can be obtained by the following equation:

$$\tau_f = \frac{f_{85} - f_{25}}{f_{25}(85 - 25)} \times 10^6 (\text{ppm}/^\circ\text{C}) \quad (1)$$

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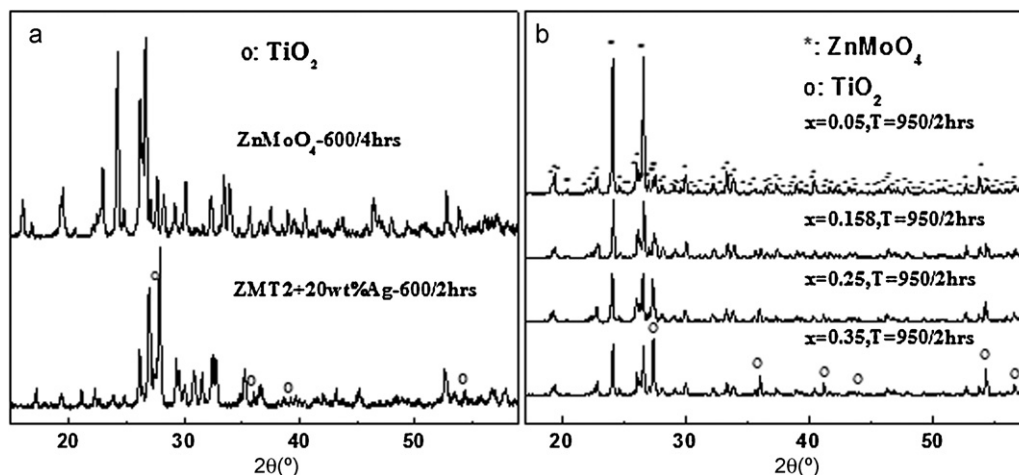


Fig. 1. XRD patterns of calcined ZnMoO_4 powders, ZMT2 cofired with 20 wt% Ag and sintered $(1-x)\text{ZnMoO}_4-x\text{TiO}_2$ composites (*, triclinic ZnMoO_4 phase; O, tetragonal rutile phase).

where f_{85} and f_{25} were the resonant frequencies at 85°C and 25°C , respectively.

3. Results and discussions

Fig. 1(a) shows the X-ray diffraction patterns of calcined ZnMoO_4 powders, $\text{ZnMoO}_4\text{-TiO}_2$ composites cofired with 20 wt% Ag and Fig. 1(b) shows the XRD patterns of $(1-x)\text{ZnMoO}_4-x\text{TiO}_2$ mixed phases sintered at 950°C . The ZnMoO_4 with triclinic wolframate structure was obtained when calcined at 600°C for 4 h. From Fig. 1(b), it can be concluded that TiO_2 coexists with ZnMoO_4 . When the content of TiO_2 increasing, the intensity of the reflections of TiO_2 increases greatly. Finally, a study of the chemical compatibility of $\text{ZnMoO}_4\text{-TiO}_2$ compounds with Ag powders has been made. As shown in Fig. 1(a), Ag can form the $\text{Ag}_2\text{Zn}_2(\text{MoO}_4)_3$ phase with ZnMoO_4 easily, and seems not to react with TiO_2 .

Fig. 2 presents the backscattered electron image of the surface of $0.65\text{ZnMoO}_4\text{-}0.35\text{TiO}_2$ sample. From the micrograph, it is seen that there are two types of grains in the specimen and the grain size is in the range of $1\text{-}6\ \mu\text{m}$. The EDS analysis shows that the light grains as B belong to ZnMoO_4 phase and the dark ones as A belong to the TiO_2 phase. Since the sintering temperature of pure TiO_2 ceramic is about 1300°C [10], the TiO_2 grains in the compounds are smaller than those of pure TiO_2 phase.

Fig. 3(a) presents the densities of the $(1-x)\text{ZnMoO}_4-x\text{TiO}_2$ ceramics. It can be seen that the densities of all the samples are larger than $4.06\ \text{g/cm}^3$, which indicates that all the specimens' relative densities are over 94%. Therefore, all the composite ceramics are well densified below 950°C . The microwave dielectric properties of the $(1-x)\text{ZnMoO}_4-x\text{TiO}_2$ ceramics are shown in Fig. 3(b) and (c). In the composite, the permittivity is determined by the permittivity, volume fraction and the complex form of the composing material [19]. If the composite material is aligned parallel to the electric field, the ε is given as [19]

$$\varepsilon = y_1\varepsilon_1 + y_2\varepsilon_2 \quad (2)$$

where ε_1 and ε_2 are permittivities of material 1 and material 2, respectively; y_1 and y_2 are the volume fractions. When the composite material is aligned in series with the electric field, the ε can be obtained as follows [19]

$$\frac{1}{\varepsilon} = \frac{y_1}{\varepsilon_1} + \frac{y_2}{\varepsilon_2} \quad (3)$$

In case the composite material is aligned randomly, the ε follows the empirical logarithmic rule [19]

$$\log \varepsilon = y_1 \log \varepsilon_1 + y_2 \log \varepsilon_2 \quad (4)$$

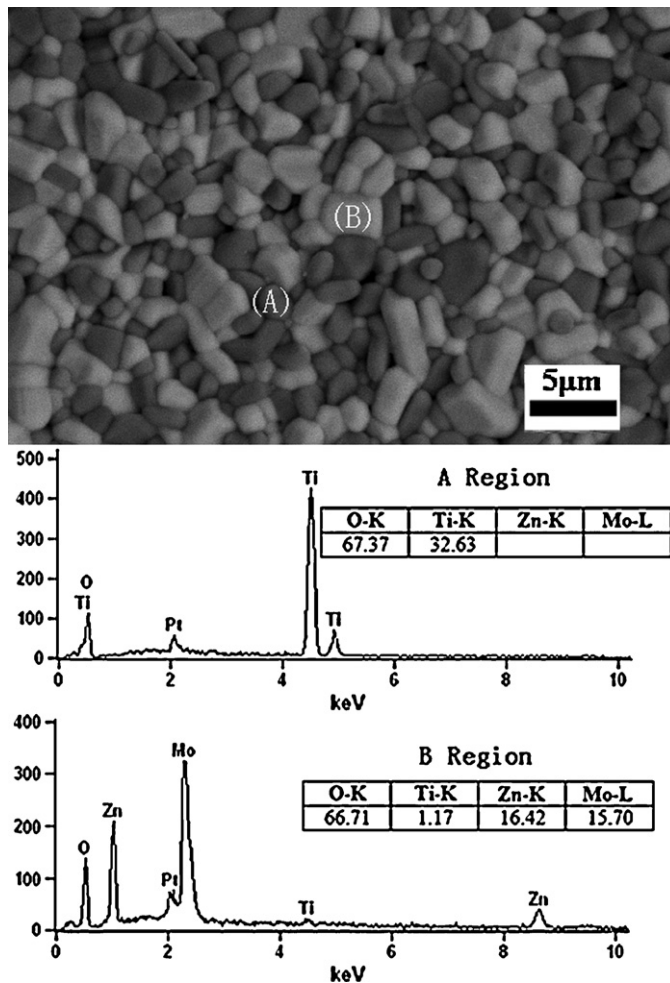


Fig. 2. The backscattered electron micrograph and EDS analysis of $0.65\text{ZnMoO}_4\text{-}0.35\text{TiO}_2$ specimen sintered at 975°C for 2 h: (a) A region, TiO_2 and (b) B region, ZnMoO_4 .

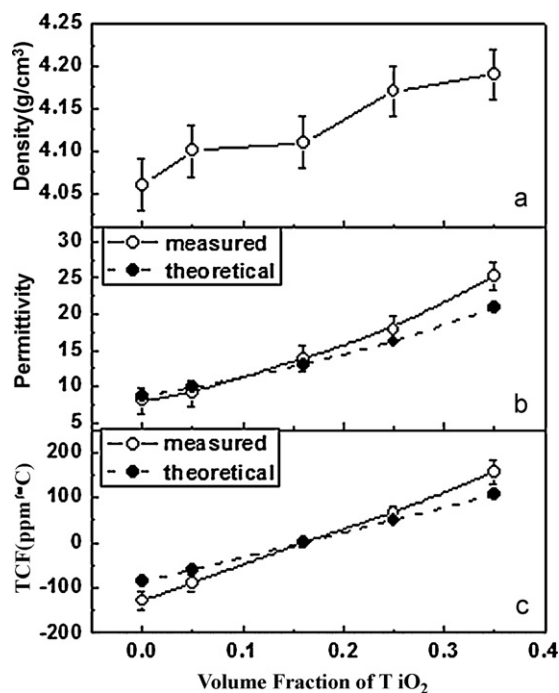


Fig. 3. Densities and microwave dielectric properties of $(1-x)\text{ZnMoO}_4-x\text{TiO}_2$ compounds sintered at their optimal temperatures for 2 h with $x=0, 0.05, 0.158, 0.25$, and 0.35 .

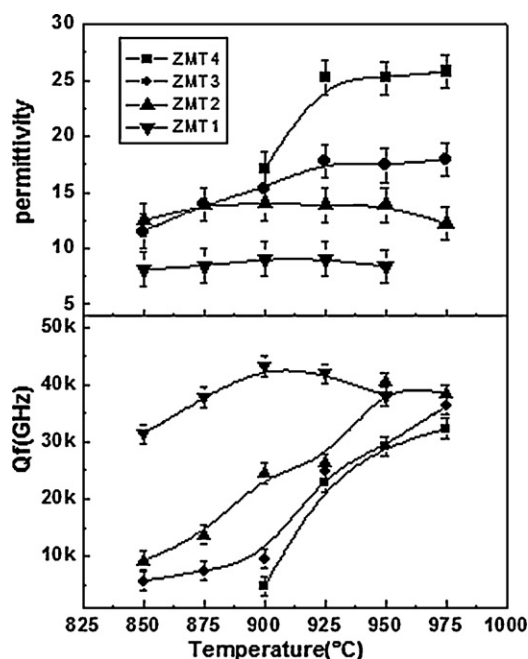


Fig. 4. Microwave dielectric properties of $(1-x)\text{ZnMoO}_4-x\text{TiO}_2$ ceramics as a function of sintering temperature.

In the $(1-x)\text{ZnMoO}_4-x\text{TiO}_2$ ceramics, the triclinic wolframite ZnMoO_4 phase and tetragonal rutile TiO_2 phase are distributed randomly. So the theoretical permittivity of the mixed ceramic can be gained by logarithmic rule. Fig. 3(b) shows that the permittivity of the composite ceramic increases from 8.0 to 25.2 when the volume fraction of TiO_2 increasing from 0 to 0.35. Although the measured permittivity is a little higher than the theoretical value, it generally accords to the logarithmic rule.

According to the empirical logarithmic rule, the τ_f of the mixed ceramic can be obtained as follows: [13]

$$\tau_f = y_1 \tau_{f1} + y_2 \tau_{f2} \quad (5)$$

where τ_{f1} and τ_{f2} are the τ_f values of material 1 and material 2, respectively. The theoretical and measured τ_f values are plotted in Fig. 3(c). The measured τ_f value shifts from -128.9 to 157.4 ppm/°C as x value increasing from 0 to 0.35. Compared with the previous reports [6,18], the measured τ_f value of pure ZnMoO_4 is a little lower and the measured τ_f value of rutile TiO_2 is a little larger, which may be caused by the different calcination temperatures. The τ_f values of $(1-x)\text{ZnMoO}_4-x\text{TiO}_2$ ceramics agree well with the mixture function (5) and the near-zero τ_f value is obtained at $x=0.158$.

The permittivities and Q_f values of the composite ceramics as a function of sintering temperature are plotted in Fig. 4. It is seen that the $(1-x)\text{ZnMoO}_4-x\text{TiO}_2$ ceramics have good Q_f values at low temperatures. The $0.35\text{ZnMoO}_4-0.65\text{TiO}_2$ composite has a minimum Q_f value (32,300 GHz) and $0.05\text{ZnMoO}_4-0.95\text{TiO}_2$ composite has a maximum Q_f value (43,300 GHz). The $0.842\text{ZnMoO}_4-0.158\text{TiO}_2$ composite with a near-zero τ_f possesses a Q_f value about 40,400 GHz.

4. Conclusion

This study introduced a new dielectric material system $(1-x)\text{ZnMoO}_4-x\text{TiO}_2$ with good microwave dielectric properties. All the samples can be well densified below 950°C . The X-ray analysis reveals that the triclinic wolframite ZnMoO_4 phase can coexist with the tetragonal rutile TiO_2 phase. Silver reacts with ZnMoO_4 easily to form the $\text{Ag}_2\text{Zn}_2(\text{MoO}_4)_3$ phase, and seems not to react with TiO_2 . When the volume fraction of TiO_2 was 0.158, the ϵ_r , Q_f value, and τ_f value of the compound were 13.9, 40,400 GHz, and $+2.0$ ppm/°C, respectively.

Acknowledgements

This work was supported by the National 973-project of China (2009CB623302), NSFC projects of China (61025002 and 109790365) and National Project of International Science and Technology Collaboration (2009DFA51820).

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