

# Microstructure and Transparent Super-Hydrophobic Performance of Vacuum Cold-Sprayed $\text{Al}_2\text{O}_3$ and $\text{SiO}_2$ Aerogel Composite Coating

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**Abstract** In this study, vacuum cold spraying was used as a simple and fast way to prepare transparent super-hydrophobic coatings. Submicrometer-sized  $\text{Al}_2\text{O}_3$  powder modified by 1,1,2,2-tetrahydroperfluorodecyltriethoxysilane and mixed with hydrophobic  $\text{SiO}_2$  aerogel was employed for the coating deposition. The deposition mechanisms of pure  $\text{Al}_2\text{O}_3$  powder and  $\text{Al}_2\text{O}_3$ - $\text{SiO}_2$  mixed powder were examined, and the effects of powder structure on the hydrophobicity and light transmittance of the coatings were evaluated. The results showed that appropriate contents of  $\text{SiO}_2$  aerogel in the mixed powder could provide sufficient cushioning to the deposition of submicrometer  $\text{Al}_2\text{O}_3$  powder during spraying. The prepared composite coating surface showed rough structures with a large number of submicrometer convex deposited particles,

characterized by being super-hydrophobic. Also, the transmittance of the obtained coating was higher than 80% in the range of visible light.

**Keywords** super-hydrophobicity · transparency · alumina · silica aerogel · vacuum cold spray

## Introduction

Super-hydrophobic solid surfaces with water contact angle larger than  $150^\circ$  and sliding angle lower than  $10^\circ$  have received widespread attention over the years (Ref 1, 2). Initially, the super-hydrophobic phenomenon has been discovered through the natural surfaces, such as plant leaves (e.g., lotus leaves, rice leaves, and rose petals) and some parts of the animal bodies (e.g., butterfly wings and water strider's legs) (Ref 3, 4). These studies revealed that surface structure and surface free energy are the two factors determining the surface wettability, and surfaces with super-hydrophobic properties require low surface free energy and suitable rough structure morphologies (Ref 5, 6).

With the continuous development of industrial production, increasing attention has been paid to better understanding and studying of nature. Materials characterized by only one main functionality can hardly meet current requirements, so providing wide ranges of performances can broaden their applications. This initiated research on transparent super-hydrophobic surfaces, combining both super-hydrophobicity and transparency of surfaces. Super-hydrophobicity reduces the adhesion of water droplets on the surface; hence, the droplets can easily roll off and remove contaminant particles along the process, keeping the surface clean and dry (Ref 7). Therefore, transparent

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super-hydrophobic surfaces have a wider range of potential applications. They may be used as self-cleaning glass, in building materials, electronic equipment screen, aircraft, cars, and spacecraft windshield glasses (Ref 8, 9).

To date, a large number of preparation methods of transparent super-hydrophobic surfaces have been developed, including *in situ* deposition (Ref 10), chemical vapor deposition (Ref 11), layer-by-layer assembly method (Ref 12), and etching (Ref 13). However, these methods suffer from numerous shortcomings, such as multiple step preparation, extreme conditions, and use of special instruments. In recent years, spray coating methods have been used for the preparation of transparent super-hydrophobic surfaces, including liquid flame spraying, spray pyrolysis technology, and suspension spraying. Using spray process, various materials (e.g., organic matter, oxides, and carbon nanotubes) can be deposited onto any substrate surface (e.g., glass, metal, and paper) (Ref 14–19). Due to their advantages in terms of simplicity, efficiency, and low-energy consumption, spray methods are considered promising for preparing transparent super-hydrophobic surfaces at large scale.

Vacuum cold spraying is one of the advanced simple and efficient spray technologies, mainly used for the preparation of ceramic coatings. The method is based on initial solid ceramic powder impacting the substrate at a certain speed to form coatings (Ref 20). Relevant reports showed that this method can deposit coatings with rough surface structures and good transparency. Hence, vacuum cold spraying is expected to yield transparent super-hydrophobic coatings (Ref 21, 22). Ceramics are widely used as functional materials, and thanks to their high hardness characteristic, it is possible to effectively improve the durability of the resulting coatings. Generally, the preparation of ceramic components requires high sintering temperatures. The joining of ceramic coatings with metals, glasses, and other materials with low melting points is usually difficult (Ref 23). Ceramic coatings produced by the conventional cold spraying method are often defective due to the poor deformation of brittle ceramic materials. Particles with high speeds will erode the substrate and the already deposited coatings (Ref 24). The emergence of vacuum cold spraying technology solves many of the above-mentioned limitations. The spraying conditions, including low absolute pressure, small starting particle size, and absence of gas preheating, induce particles with relatively low velocity and low temperature when impacting the substrate. This allows the particles to effectively deposit on the substrate. The technique has the ability to produce ceramic coatings with good performances without using high-temperature condition processing to coat almost any substrate (Ref 25). Moreover, many spray prepared coatings still have problems in terms

of bond strength. Using vacuum cold spraying, a portion of the kinetic energy of the particles dissipated during collision with the substrate is converted to binding energy. Mechanical bonding can take place between the substrate-particle and particle-particle interfaces to form coatings with relatively high strength and adhesion, suitable for practical applications (Ref 23, 24).

In this study, vacuum cold spraying was used to deposit transparent super-hydrophobic coatings on glass substrates. Two common categories of materials used for preparing transparent components were selected as the deposited powders (Ref 9, 26). One of the powders was submicrometer  $\text{Al}_2\text{O}_3$ , the most widely used ceramic material. Because of its low price, the production cost will greatly reduce. Another powder material was loose porous  $\text{SiO}_2$  aerogel added to  $\text{Al}_2\text{O}_3$  powder during spraying to regulate the structures of the composite coating surfaces. This study aimed to discuss the deposition mechanisms of the powders with different structures, as well as the effects of powder on the structure, hydrophobicity, and transparency of the resulting coatings. The traditional method for preparing super-hydrophobic surface using hydrophilic materials aimed mainly to form surfaces with rough structures and then use low surface free energy materials to modify the surfaces (Ref 27). However, the modification reaction may not be sufficient and the modified layer could easily be damaged. The present research subverted the traditional preparation process, attempting to spray super-hydrophobic coatings directly by the materials with hydrophobic characteristics. Since  $\text{Al}_2\text{O}_3$  has intrinsic hydrophilicity, first the powder was modified with fluorosilane to transfer to the hydrophobicity, while commercial  $\text{SiO}_2$  aerogel powder had initial intrinsic hydrophobicity. The study discussed the feasibility of using hydrophobic materials to prepare super-hydrophobic surfaces with desirable low surface free energy. The proposed study may provide an innovative approach to prepare transparent super-hydrophobic surfaces through the vacuum cold spraying method.

## Experimental

### Materials

Two kinds of materials,  $\text{Al}_2\text{O}_3$  (AL-160SG-3, Showa Denko, Japan) and  $\text{SiO}_2$  aerogel (AEROSIL R106, Degussa, Germany), were used for spraying. The morphologies of the powders are shown in Fig. 1. The particle size of  $\text{Al}_2\text{O}_3$  powder was in the submicrometer grade with size distribution of 200–500 nm. The particles showed angular morphologies and slight agglomeration (Fig. 1(a)). The commercially available  $\text{SiO}_2$  aerogel was an intrinsic hydrophobic material. As shown in Fig. 1(b), the powder

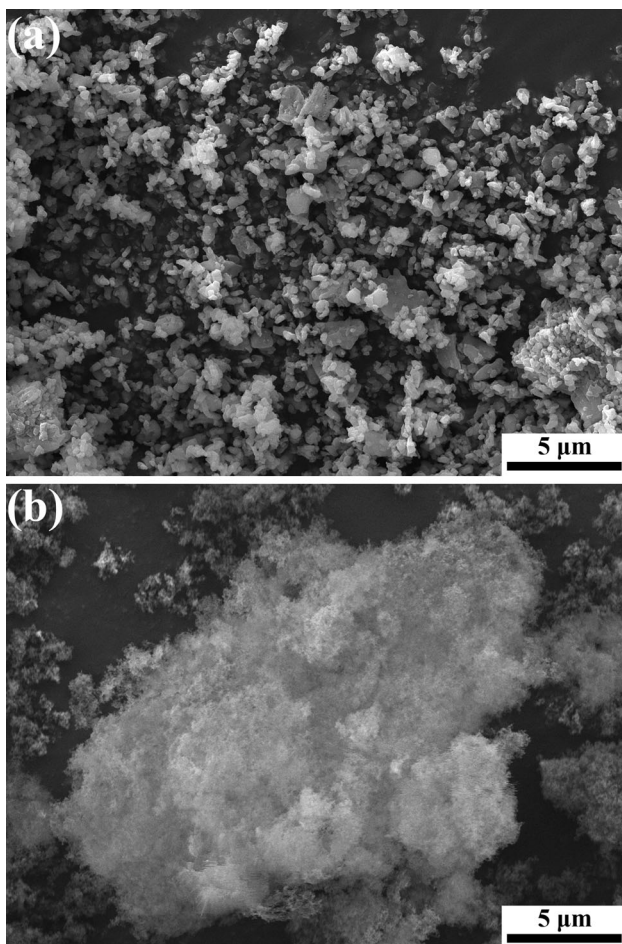
was composed entirely of nanoparticles with an average size of 10 nm and exhibited soft agglomerated structure with micrometer sizes.

Commercially available glass slides (No. 7101, Sail Brand, China) with sizes of  $75 \times 25 \times 1$  mm were used as substrates. Before spraying, the substrates were cleaned with acetone and alcohol, respectively. Glass was selected because of its wide potential applications and high transparency to facilitate the detection of the transmittance of the coating.

The low surface energy material used in the experiment was 1,1,2,2-tetrahydroperfluorodecyltriethoxysilane (F-1060, SICONG, China), referred to as fluorosilane (FAS) with the chemical formula of  $\text{CF}_3(\text{CF}_2)_7\text{CH}_2\text{CH}_2\text{Si}(\text{OCH}_2\text{CH}_3)_3$ .

### Powder Modification

Since  $\text{Al}_2\text{O}_3$  had an intrinsic hydrophilicity, the powder was modified with fluorosilane to reduce its surface free



**Fig. 1** Surface morphologies of the powders. (a) The submicrometer  $\text{Al}_2\text{O}_3$  powder with particle sizes of 200–500 nm and (b)  $\text{SiO}_2$  aerogel with soft agglomerates composed of large numbers of nanoparticles

energy. The procedure of the powder modification was performed as follows: 95 ml ethanol and 5 ml fluorosilane were first mixed to form a solution. The pH was adjusted to 11 using aqueous ammonia, and the mixture was ultrasonically stirred for 1 hour. Second, appropriate amount of  $\text{Al}_2\text{O}_3$  powder (about 100 g) was mixed with 250 ml ethanol and stirred for 10 min to obtain a suspension. Next, the fluorosilane solution was added dropwise to the  $\text{Al}_2\text{O}_3$  suspension and stirred for 18 h at room temperature using a magnetic stirrer, and the resulting powder was filtered off and then dried at 80 °C.

### Low-Energy Ball Milling

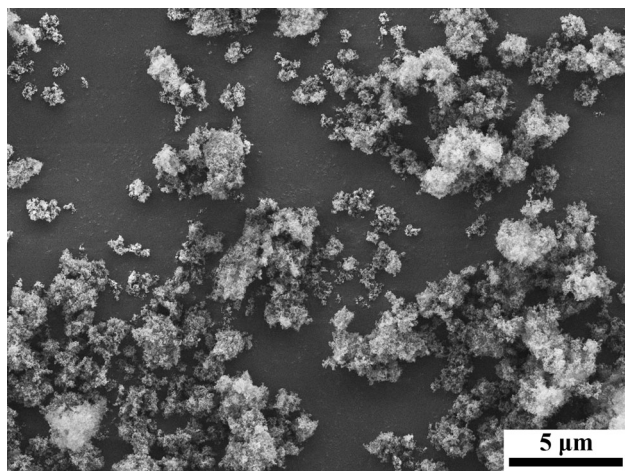
The previous literature showed that super-hydrophobic surfaces often had hierarchically rough structures (Ref 6). To yield suitable surface structures and study how the powder structure affect the coating structures and properties using vacuum cold spraying, hydrophobic  $\text{SiO}_2$  aerogel was added to modified  $\text{Al}_2\text{O}_3$  powder and mixed by low-energy ball milling. The parameters of low-energy ball milling are given in Table 1.  $\text{Al}_2\text{O}_3$  powder and  $\text{SiO}_2$  aerogel with four mass ratios were used for milling. The milling media was  $\text{Al}_2\text{O}_3$  grinding ball with a diameter of 3 mm. Figure 2 shows typical morphology of the powder when  $\text{Al}_2\text{O}_3$  particles were mixed with  $\text{SiO}_2$  aerogel at the mass ratio of 3:1. The submicrometer particles were closely adhered by large numbers of nanoparticles. The mixed powder exhibited micrometer-sized cluster-like structures.

### Coating Preparation

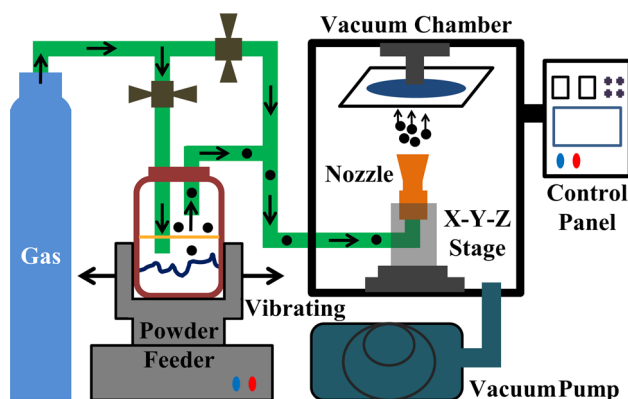
A home-developed vacuum cold spraying (VCS) system by Xi'an Jiaotong University was employed to deposit the modified  $\text{Al}_2\text{O}_3$  and mixed powders (Fig. 3). During spraying, helium (He) was used as both the delivering powder and accelerating gas. The delivery powder gas entered the aerosolized feeder through the conduit. The particles were then effectively sent into the gun using the gas flow thanks to vibration of the feeder. With the accelerating gas, the particles would hit the substrate at certain speeds during deposition. The substrate was kept immobile, while the gun scanned in the X-Y plane and the powder sprayed evenly over the substrate surface.

**Table 1** The low-energy ball milling parameters

Parameter	Value
Milling rotational speed ( $\text{r min}^{-1}$ )	180
Ball milling time (h)	24
Mass ratio of ball to powder	10:1
Mass ratio of $\text{Al}_2\text{O}_3$ powder to $\text{SiO}_2$ aerogel	5:1/4:1/3:1/2:1



**Fig. 2** Morphology of the mixed powder. The mass ratio of  $\text{Al}_2\text{O}_3$  to  $\text{SiO}_2$  is 3:1



**Fig. 3** Schematic diagram of the vacuum cold spraying system

Scanning the entire surface was considered as a 1 time spray process. In this study, rectangular copper with hollow size of  $0.25 \text{ mm} \times 2.5 \text{ mm}$  was used as the nozzle. Table 2 gathers the parameters used for vacuum cold spraying.

### Characterization

Morphologies of feedstock powders and coatings were characterized by field emission scanning electron microscopy (MIRA3 LMH, TESCAN, Czech Republic). Surface roughness ( $R_a$ ) was detected by laser scanning confocal microscopy (LSCM, VK9700K, Keyence, Japan). The static water contact angle (CA) and sliding angle (SA) measurements were carried out using a contact angle instrument (JC2000C4, Shanghai Zhongchen Digital Technic Apparatus Co., Ltd., China). The volume of the freshly deionized water droplet used for measurements was  $5 \mu\text{L}$ . The static contact angle was measured using the sessile drop method by putting a water droplet on the

**Table 2** Parameters of the vacuum cold spraying

Parameter	Unit	Value
Gas (He) flow rate	$\text{L min}^{-1}$	6
Chamber pressure	Pa	< 200
Gun traveling speed	$\text{mm s}^{-1}$	10
Spraying distance	mm	15
Times of spraying	Time	2

horizontal coating surface followed by capturing an image and calculating the contact angle from the image. The sliding angle was measured by putting a water droplet on the horizontal coating surface followed by tilting the coating until the droplet started to move on the surface. The inclination angle of the coating was considered as the sliding angle. At least 5 droplets measurements were employed for each sample at different locations.

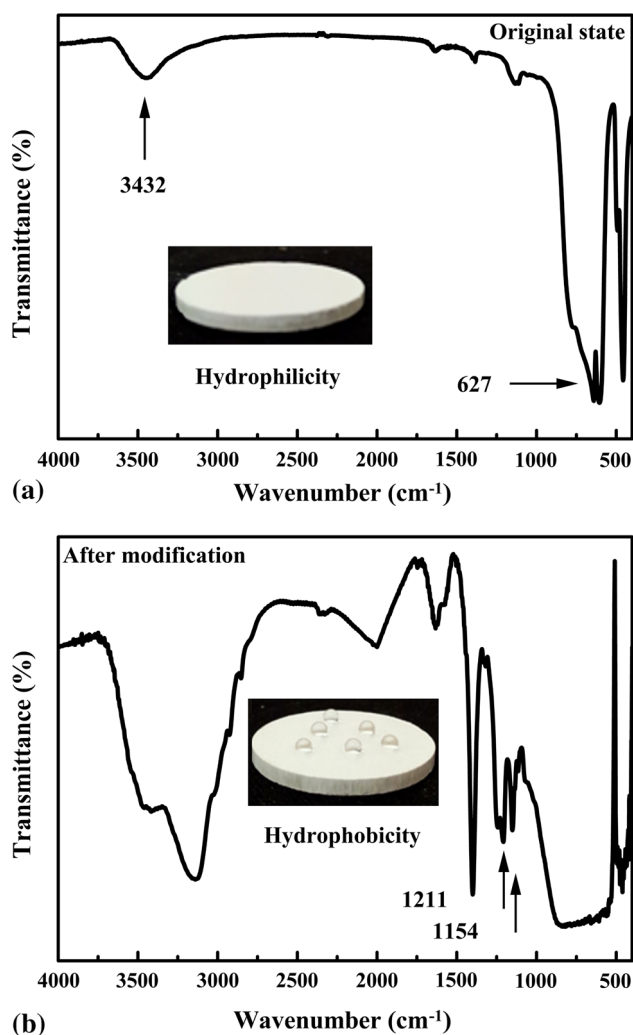
The chemical composition of the powder was detected by Fourier transform infrared spectroscopy (FTIR, Vertex70, Bruck, Germany) with a resolution of  $4 \text{ cm}^{-1}$  and 64 scans. The transmittance values of the coatings were collected with an ultraviolet-visible spectrophotometer (UV-3600, Shimadzu, Japan) in the range of 380–760 nm wavelengths. Soaking method and ultrasonic cleaning method were utilized for stability testing of the transparent super-hydrophobic coatings. The coating was first placed in alcohol at liquid depth of 6 mm, and using test time as a variable, the corresponding water contact angle and sliding angle on the surface were recorded after each test.

## Results and Discussion

### Characterization of Modified $\text{Al}_2\text{O}_3$ Powder

The infrared spectra of  $\text{Al}_2\text{O}_3$  powders before and after modification by fluorosilane are shown in Fig. 4. The chemical groups associated with hydroxyl and oxide were detected on the original powder surface, where the transmission peak around  $3432 \text{ cm}^{-1}$  was assigned to the stretching vibration of O-H and that at  $627 \text{ cm}^{-1}$  represented the oxide (Ref 28). Since the surface free energies of hydroxyl and metal oxide are greater than that of water, the original powder exhibited hydrophilicity (Ref 29). It can be observed that the water droplets can rapidly penetrate into the tablet surface after the original powder was tabletted (inset image in Fig. 4(a)).

For the modified powder, water droplets remained standing on the surface and exhibited spherical shapes after the powder was tabletted (inset image in Fig. 4(b)), indicating that the powder was hydrophobic. The FTIR data



**Fig. 4** Comparison of infrared spectra of the powders. (a) Original powder and (b) modified powder. Inserted images show the droplet states on the powders tablet surfaces

showed that the asymmetric stretching vibration and symmetrical stretching vibration of  $-CF_2$  were located at 1211 and 1154  $cm^{-1}$ , further confirming that the particle surface was grafted with fluorosilane molecules with low surface free energy (Ref 30). The mechanism of powder modification process is presented in Fig. 5. First, the fluorosilane was hydrolyzed by replacing the three ethoxy groups attached to the Si atom by the hydroxyl groups. Afterward, the hydroxyl groups present on both the fluorosilane and particle surface underwent dehydration condensation, where low surface energy fluorocarbon chains were grafted onto the particle surface (Ref 31). Meanwhile, dehydration reaction occurred between the hydrolyzed fluorosilane molecules, forming a network of fluorosilane molecular chain on the surface. During this process, the powder particle surface was expected to sufficiently be covered by low surface free energy molecules. In addition, it will be

noted that the modification process did not change the morphology of the original  $Al_2O_3$  powder as revealed by SEM. In other words, the morphology of the modified powder was basically the same as that of the powder shown in Fig. 1(a).

### Coating Deposition Using Modified $Al_2O_3$ Powder

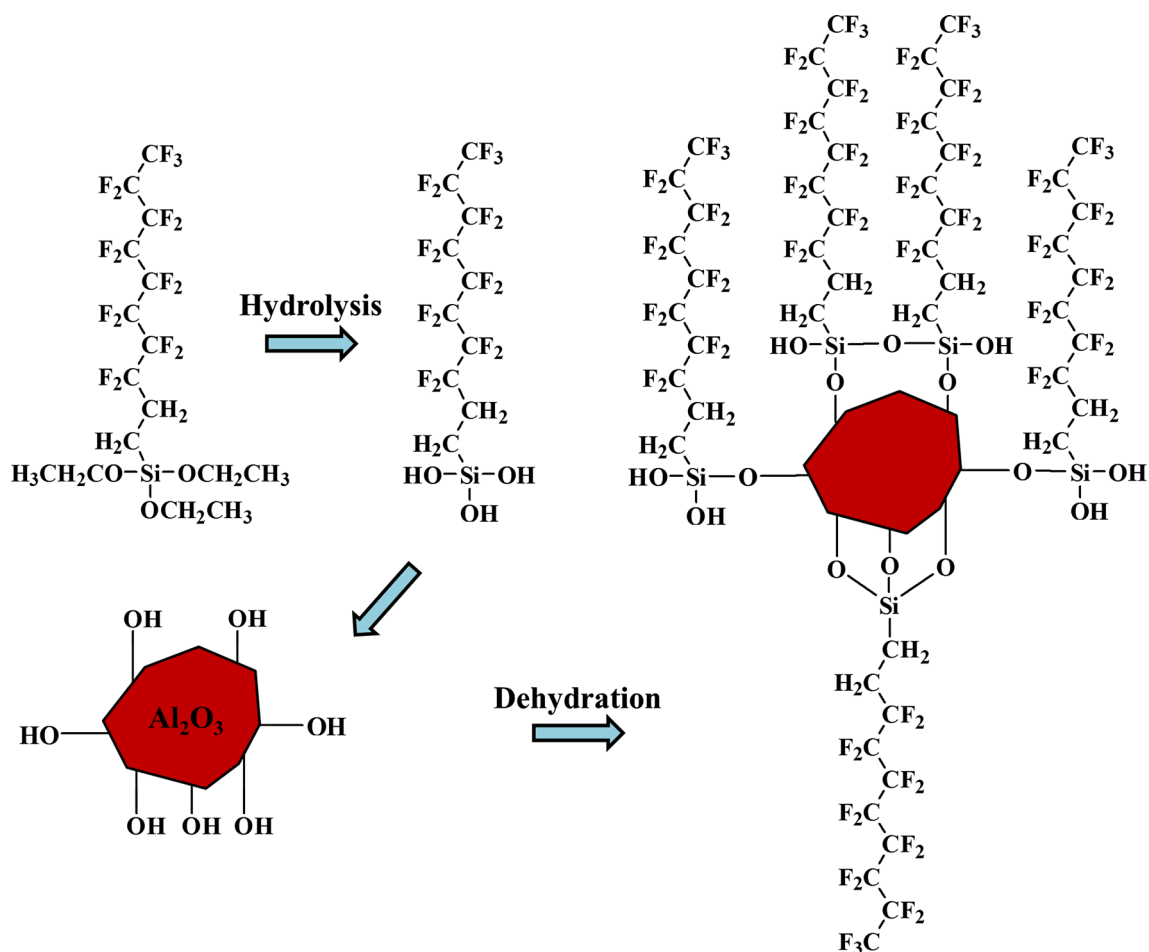
First, the modified  $Al_2O_3$  powder was used to deposit coatings on glass substrates by vacuum cold spraying. Figure 6 shows the morphology of the coating surface under different magnifications. The particles were uniformly deposited on the substrate surface in a flattened manner to form a flat surface. The average roughness (Ra) of the surface detected by LSCM was  $0.195 \pm 0.008 \mu m$ .

Figure 7 depicts the light transmittance of the glass substrate and deposited coating. It can be seen that the transmittance in the visible range of  $Al_2O_3$  coating reached a value about 95%, almost the same as the glass substrate. The deposited coating illustrated good light transmittance. Some studies found that  $Al_2O_3$  is an intrinsic transparent material since both its grain boundary reflection and scattering losses could be ignored. Therefore,  $Al_2O_3$  was commonly used in the preparation of transparent components (Ref 32, 33). The good light transmittance of the coating indicated small surface roughness and high density of the coating, which greatly inhibited light scattering.

The evaluation of surface wettability revealed the smooth glass substrate surface to be hydrophilic with water contact angle of only  $23.6^\circ \pm 3.2^\circ$ . When  $Al_2O_3$  coating was deposited on the substrate, the water contact angle value increased to  $107.1^\circ \pm 1.7^\circ$ . An image of the water droplet on the surface is shown in Fig. 6(a). The deposited  $Al_2O_3$  coating surface showed hydrophobicity, indicating that the surface chemical state of the powder did not change and its hydrophobic characteristic was retained during spraying. On the other hand, the hydrophobic coating surface can be obtained by direct deposition of the modified powder. However, since water droplets cannot roll on the surface and pinned to it, the surface failed to gain super-hydrophobicity. This was presumably caused by the flat surface structure and low surface roughness. Therefore, appropriate surface roughness was necessary for excellent hydrophobicity of the coating surface.

### Coatings Deposition Using Mixed Powders

$SiO_2$  is a low light-absorbing material commonly used for preparing transparent components, similar to  $Al_2O_3$  (Ref 17, 34). To induce relevant structures in terms of super-hydrophobicity and transparency of the coating, hydrophobic  $SiO_2$  aerogel with various proportions was added to the modified  $Al_2O_3$  powder to deposit the coating.



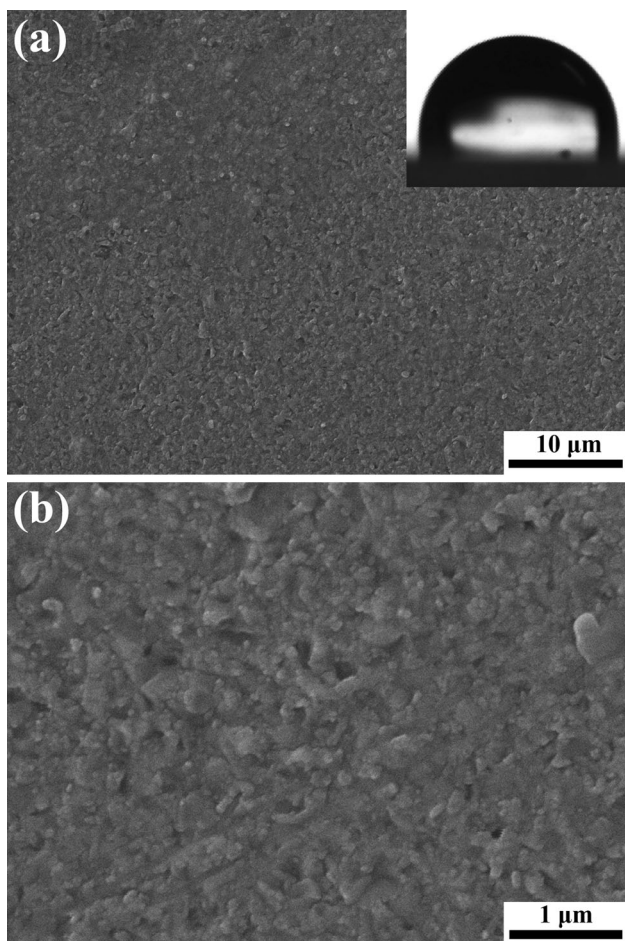
**Fig. 5** Schematic representation of the powder modification process

The effect of powder structure on the coating structure and its properties were then evaluated. Four sets of mixed Al<sub>2</sub>O<sub>3</sub> powder and SiO<sub>2</sub> aerogel with different mass ratios are listed in Table 1 and were prepared by low-energy ball milling.

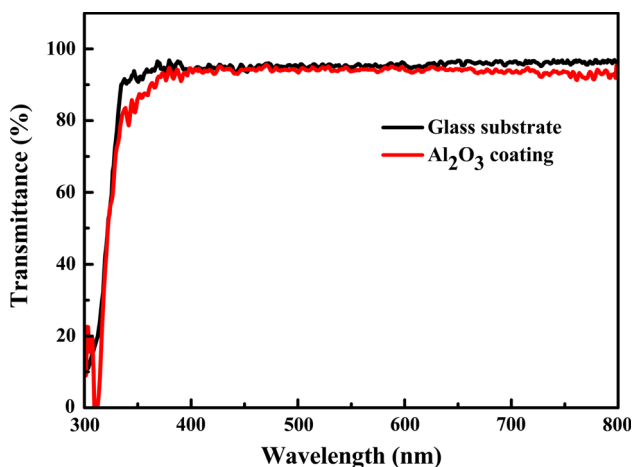
Compared to the coating surface deposited by single Al<sub>2</sub>O<sub>3</sub> powder, the surface wettability showed improvements in surface hydrophobicity when SiO<sub>2</sub> aerogel was added to Al<sub>2</sub>O<sub>3</sub> powder (Fig. 8). As the proportion of SiO<sub>2</sub> aerogel increased in the mixed powder, the coating surface exhibited different hydrophobicities. At mass ratios of Al<sub>2</sub>O<sub>3</sub> powder and SiO<sub>2</sub> aerogel of 5:1 and 4:1, the contact angle of water droplets on coating surfaces raised to  $129.2^\circ \pm 1.7^\circ$  and  $135.5^\circ \pm 1.5^\circ$ , respectively. Unfortunately, water droplets still pinned on the surfaces and cannot roll freely, showing hydrophobic surfaces. When the mass ratio reached 3:1, the contact angle of the droplet on the surface rose to  $153.8^\circ \pm 0.9^\circ$ . In addition, the droplet can quickly roll off from the surface when the coating was tilted by  $2^\circ$ . In this case, the prepared coating surface was super-hydrophobic. The continuous increase in SiO<sub>2</sub>

aerogel content in the mixed powder to reach a mass ratio of 2:1 reduced the water contact angle to  $145.5^\circ \pm 1.2^\circ$  and raised the sliding angle to  $16.0^\circ \pm 0.5^\circ$ .

The addition of SiO<sub>2</sub> aerogel into the Al<sub>2</sub>O<sub>3</sub> powder also affected the light transmittance of the coatings. Figure 9 shows the light transmission of the composite coatings. As the content of SiO<sub>2</sub> aerogel in the mixed powder increased, the light transmittance of the coating gradually decreased. The coatings prepared by mixed powders with mass ratios of Al<sub>2</sub>O<sub>3</sub> powder to SiO<sub>2</sub> aerogel of 5:1 and 4:1 showed excellent transparencies with light transmittance near 90% in the range of visible light. At mass ratio of 3:1, the light transmission of the coating decreased, but the coating was still transparent with light transmittance above 80%. However, the light transmittance of the coating deposited by powder at mixing ratio of 2:1 was only 70%–80%, showing translucent properties. Overall, the above data revealed that the coating prepared by mixed powder at ratio of 3:1 was super-hydrophobic with good light transmittance.

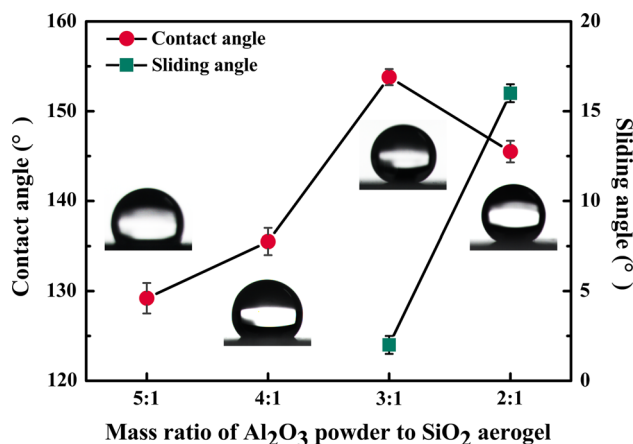


**Fig. 6** Morphology of the coating surface deposited by modified  $\text{Al}_2\text{O}_3$  powder under different magnifications. The inset image shows the water droplet standing on the coating surface with static contact angle of  $107.1^\circ$

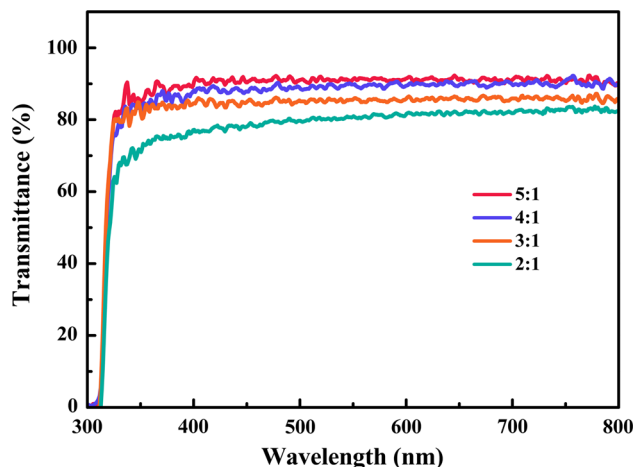


**Fig. 7** The transmittance of the glass substrate and  $\text{Al}_2\text{O}_3$  coating

The coating structure often influences surface wettability and transmittance of the coating. The SEM images of the coating prepared by mixed powders are shown in

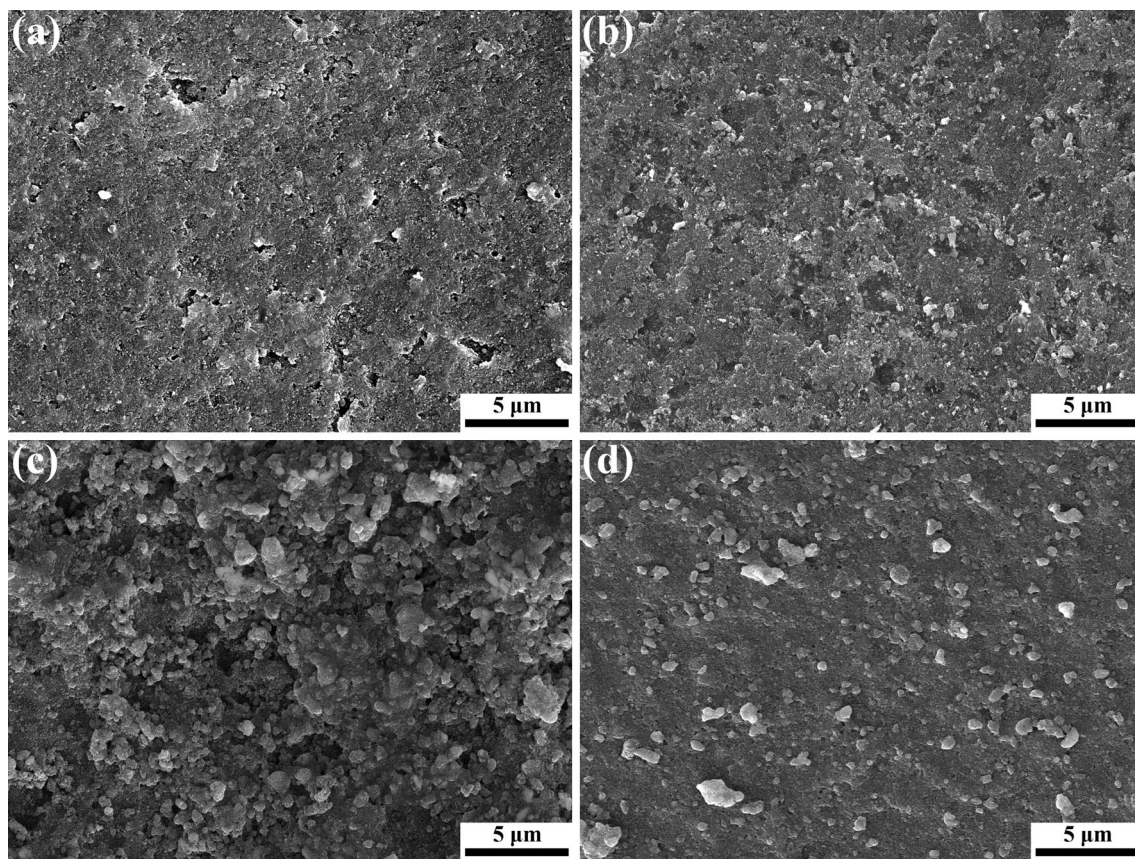


**Fig. 8** Water contact angle and sliding angle of the coating surface deposited by  $\text{Al}_2\text{O}_3$  powder and  $\text{SiO}_2$  aerogel mixed at different mass ratios



**Fig. 9** The transmittance of the composite coatings deposited by  $\text{Al}_2\text{O}_3$  powder and  $\text{SiO}_2$  aerogel mixed at different mass ratios

Fig. 10, and surface roughness values of the resulting coatings measured by LSCM are listed in Table 3. As the proportion of  $\text{SiO}_2$  aerogel in the mixed powder increased, both the surface structure and surface roughness obviously changed. At the mass ratio of  $\text{Al}_2\text{O}_3$  powder to  $\text{SiO}_2$  aerogel of 5:1 and 4:1, the coating surface remained relatively flat (Fig. 10(a, b)). However, the surface roughness of the coatings improved as compared to the surface deposited by single  $\text{Al}_2\text{O}_3$  powder. The above-mentioned coating surfaces were hydrophobic, but water droplets cannot roll on the surface, indicating that the rough surface structure preventing movement of the droplets, the liquid may fill in the depression area of the rough structure. The contact of the droplets with the surface can be regarded as in the Wenzel state, predicting that the contact angles of droplets on hydrophobic surfaces increased as surface roughness rose (Ref 35). Therefore, the increase in roughness led to improvements in surface hydrophobicity.



**Fig. 10** The coating surface deposited by  $\text{Al}_2\text{O}_3$  powder and  $\text{SiO}_2$  aerogel mixed at different mass ratios. The mass ratios of  $\text{Al}_2\text{O}_3$  powder to  $\text{SiO}_2$  aerogel are: (a) 5:1, (b) 4:1, (c) 3:1, and (d) 2:1

**Table 3** Surface roughness of the coatings

Mass ratio of $\text{Al}_2\text{O}_3$ to $\text{SiO}_2$	Ra, $\mu\text{m}$
5:1	$0.217 \pm 0.005$
4:1	$0.224 \pm 0.009$
3:1	$0.294 \pm 0.005$
2:1	$0.275 \pm 0.003$

Thus, the contact angle of the surface deposited by mixed powder was superior to that prepared by single  $\text{Al}_2\text{O}_3$ . At the mass ratio of 3:1, large numbers of submicrometer-sized particles were deposited on the surface (Fig. 10(c)), and the surface roughness of the coating reached a maximum value of  $0.294 \pm 0.005 \mu\text{m}$ . Since the coating contained undulating rough structure formed by large numbers of submicrometer particles, the recessed area formed by the rough structure can form large numbers of air cushions, which could effectively prevent the droplets from penetrating into the rough structure. In this case, the droplets were in the Cassie-Baxter state on the surface (Ref 35). This state reduced the contact area between the droplet and the surface, which greatly increased the contact angle of

the droplet and decreased the resistance of the surface to droplet movement. This yielded super-hydrophobic surfaces. However, at the ratio of 2:1, the coverage density of the submicrometer particles on the coating surface obviously decreased, as shown in Fig. 10(d). The surface roughness of the coating slightly decreased, resulting in reduced hydrophobicity.

The factors affecting the transparency of coatings are mainly the surface roughness and density, implying the introduction of sources of light scattering. Nakajima et al. (Ref 36) suggested that to maintain the transparency of materials in the visible wavelength range (380–760 nm), surface roughness should be either lower or much higher than the range. Table 3 indicates that surface roughness (Ra) values of the four different surfaces were all lower than the visible wavelength. Therefore, surface roughness of the coating was small enough to inhibit light scattering. The internal structure of the coating may be the main factor affecting the light transmittance of the prepared coatings. The light transmission results of the coatings suggested that addition of  $\text{SiO}_2$  aerogel during deposition decreased the density of the coating, which should be the reason for the declined light transmittance of the coating (Ref 37, 38).



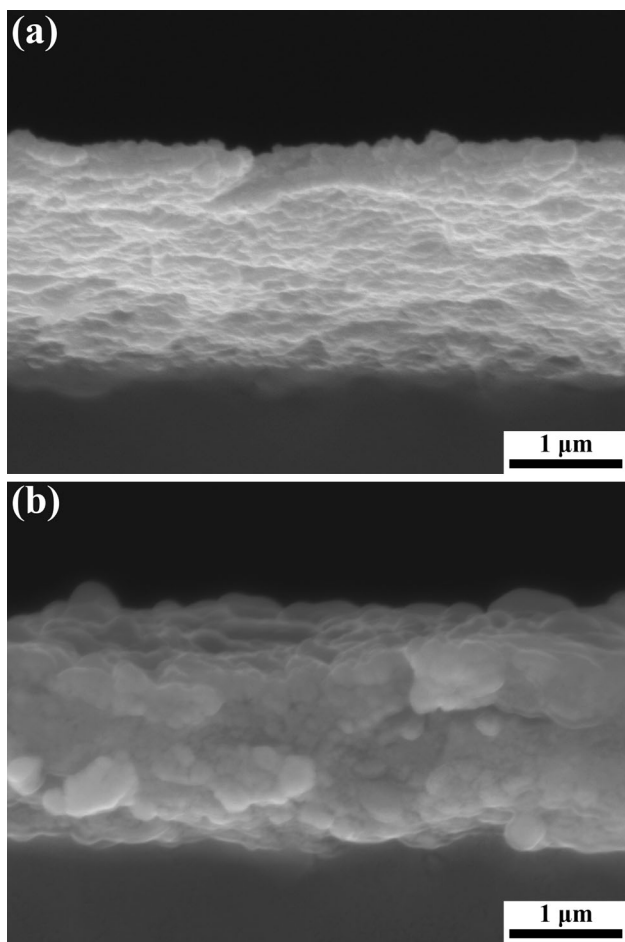
### Deposition Mechanism of Particles

The results mentioned above showed that the coating structure significantly affects their properties. By spraying either single  $\text{Al}_2\text{O}_3$  powder or mixed powders, the coating structure can effectively be controlled by adjusting the powder structure by vacuum cold spraying.

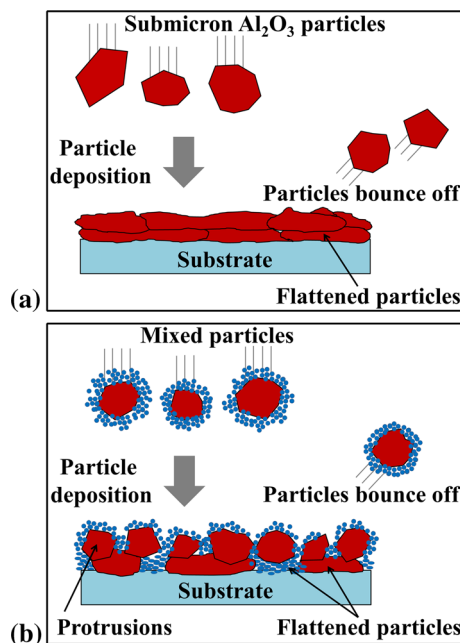
For  $\text{Al}_2\text{O}_3$  powder, the coating surface was composed of large numbers of flattened particles shown in Fig. 6. The cross-sectional morphology of the coating is depicted in Fig. 11(a). The cohesion of the particles appeared good, and the coating was very dense. The average size of the particles in the coating is smaller than that of the original submicrometer particles. The particles were also well flattened, indicating that they underwent fracture and plastic deformation at high impact velocities during spraying. These data were consistent with related published reports (Ref 25, 39). The deposition mechanism of the powder is summarized in Fig. 12(a). During particles collision with the substrate at high speed, a great force was

produced, resulting in fracture and deformation of the particles. Coupled with the hammering effect produced by subsequent bombarding particles in the deposition process, a coating with relatively flat surface morphology and dense internal structure was generated (Ref 23). Such coating structure induced excellent light transmission but poor surface hydrophobicity.

The average particle size of  $\text{SiO}_2$  aerogel was estimated to be in the nanometer scale with loose agglomerated particles (Fig. 1(b)). The density of aerogel was small, where more than 90% of powder composition was air. Therefore, large amounts of space as void existed inside the aerogel. When  $\text{SiO}_2$  aerogel was added to the submicrometer  $\text{Al}_2\text{O}_3$  powder for coating deposition at mass ratio of  $\text{Al}_2\text{O}_3$  powder to  $\text{SiO}_2$  aerogel of 3:1, the coating surface became rough and large numbers of submicrometer particles formed when compared to single alumina-deposited coating surface. Figure 11 (b) illustrates the cross-sectional morphology of the coating. The addition of  $\text{SiO}_2$  aerogel during deposition reduced the degree of fragmentation of  $\text{Al}_2\text{O}_3$  particles, inducing large amounts of bigger particles inside the coating. The morphology of mixed powder in Fig. 2 showed larger particles intimately adhered by large numbers of nano aerogel particles, forming mixed powder with micrometer-sized cluster-like structures. During the deposition process, presence of large numbers of voids in the aerogel may provide a buffering effect for deposition of submicrometer particles, thereby reducing the



**Fig. 11** Cross-sectional morphologies of the coatings. (a) Single submicrometer  $\text{Al}_2\text{O}_3$  powder and (b) mixed powder of submicrometer  $\text{Al}_2\text{O}_3$  and  $\text{SiO}_2$  aerogel at the mass ratio of 3:1



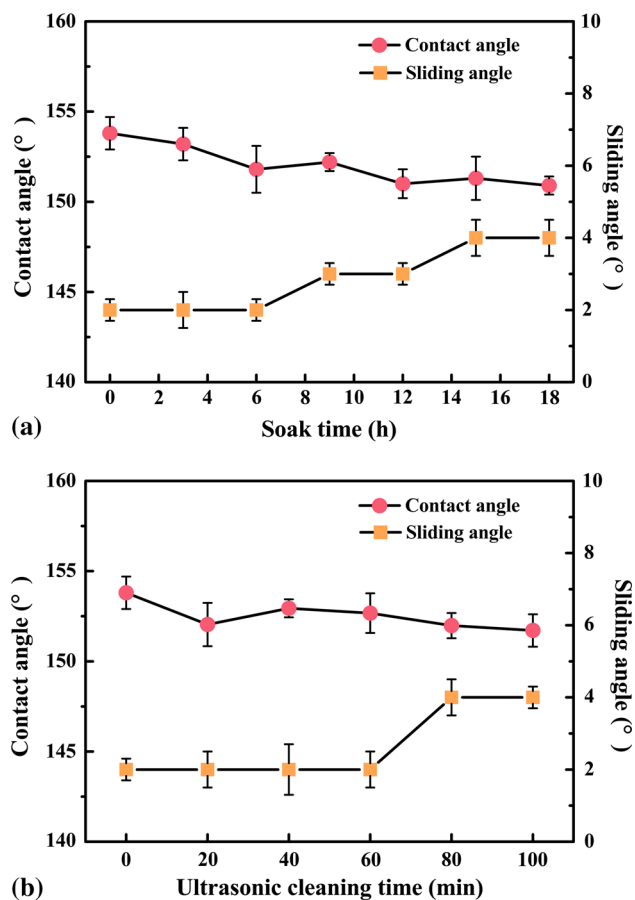
**Fig. 12** The deposition mechanism of the powders with different structures during vacuum cold spraying. (a) Single submicrometer  $\text{Al}_2\text{O}_3$  powder and (b) mixed powder of submicrometer  $\text{Al}_2\text{O}_3$  and  $\text{SiO}_2$  aerogel

fragmentation and flattening degree of the submicrometer particles. This, in turn, resulted in large numbers of bigger particles on the coating surface and inside it. The deposition mechanism is summarized in Fig. 12(b). The rough surface structure was conducive to reducing the solid-liquid contact area and promoting the surface super-hydrophobicity. However, the buffering effect of the aerogel may reduce the hammering effect of the submicrometer  $\text{Al}_2\text{O}_3$  particles on the coating, resulting in decreased density of the internal structure of the coating. This result subsequently increased light scattering of the coating and declined the transparency.

The content of silica aerogels in the mixed powder would also affect the coating structure. At smaller contents of aerogel in the mixed powder (mass ratios of  $\text{Al}_2\text{O}_3$  to  $\text{SiO}_2$  of 5:1 and 4:1), the added amount of the aerogel was too small to significantly alter the deposition behavior of the  $\text{Al}_2\text{O}_3$  particles. Therefore, the structure and properties of the coating did not significantly change when compared to that of single  $\text{Al}_2\text{O}_3$ -deposited coating. With the mixed powder with larger contents of aerogel (mass ratios of  $\text{Al}_2\text{O}_3$  to  $\text{SiO}_2$  of 2:1), the deposition was mainly based on  $\text{SiO}_2$  particles and only small amounts of submicrometer  $\text{Al}_2\text{O}_3$  particles were deposited on the surface (Fig. 10(d)). Thus, it can be speculated that the decrease in submicrometer particle content would weaken the hammering effect of the particles on the coating, decreasing the coating density, and affecting its light transmission performance. As a result, the suitable coating structures with good performances would require particles with matched sizes.

### Stability of Super-Hydrophobic Coating Surface

The stability of super-hydrophobic coating surface highly affects the practical applications. To evaluate the stability, the transparent super-hydrophobic coatings prepared with mixed powders at a mass ratio of  $\text{Al}_2\text{O}_3$  to  $\text{SiO}_2$  of 3:1 were subjected to alcohol soaking and alcohol ultrasonic cleaning treatment, respectively. The changes in contact angle and sliding angle of the coatings are shown in Fig. 13. After soaking in alcohol or ultrasonic cleaning with alcohol for some time, the wettability of the super-hydrophobic surface slightly varied. The water contact angle on the surface is still greater than  $150^\circ$ , and sliding angle is lower than  $10^\circ$ . Meanwhile, the transparency of the coating remained unchanged during the treatment. The above results indicated that the combination of the fluorosilane assembled on the coatings and the bonding of the coatings were strong enough to yield films with relevant properties. The prepared transparent super-hydrophobic surface showed certain stability. However, the adhesion strength and abrasion resistance are two important criteria for



**Fig. 13** The variation of water contact angle and sliding angle of the super-hydrophobic surface as a function of testing time. (a) Soaking in alcohol and (b) ultrasonic cleaning with alcohol

evaluating the feasibility of coatings in practical use; thus, relevant performance still requires further investigation.

### Conclusions

Vacuum cold spraying method was used to prepare transparent super-hydrophobic coatings. The results showed that deposited fluorosilane-modified submicrometer  $\text{Al}_2\text{O}_3$  surface was relatively flat and the coating was dense. The surface hydrophobicity was poor, but light transmittance was excellent. The addition of appropriate amounts of loose  $\text{SiO}_2$  aerogel to the submicrometer  $\text{Al}_2\text{O}_3$  powder during the deposition formed suitable coating structure meeting the performance requirements thanks to the aerogel which provided sufficient cushioning for the  $\text{Al}_2\text{O}_3$  deposition. At the mass ratio of  $\text{Al}_2\text{O}_3$  powder to  $\text{SiO}_2$  aerogel of 3:1, the coating surface presented rough structure with large numbers of submicrometer convex deposited particles, satisfying the conditions of super-hydrophobicity. However, the coating structure did not

cause severe light scattering, and light transmission of the coating was more than 80% in the range of visible light. When subjected to immersion or ultrasonic cleaning with alcohol, the coatings still maintained super-hydrophobicity and good light transmission. The proposed vacuum cold spraying method with simple operation and high efficiency would be promising to extend the potential applications of transparent super-hydrophobic coatings to waterproof glass, protective films, and electronic equipment screens, among others.

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